- radius from 600 to 60 nm upon acidification of suspensions of dioleoylphosphatidylcholine (2c) in aqueous PEAA solutions.
- Fichtner, F.; Schonert, H. Colloid Polym. Sci. 1977, 255, 230.
- (8) Joyce, D. E.; Kurucselv, T. Polymer 1981, 22, 415.
- Sugai, S.; Nitta, K.; Ohno, N.; Nakano, H. Colloid Polym. Sci. 1983, 261, 159.
- Chen, T. S.; Thomas, J. K. J. Polym. Sci., Polym. Chem. Ed. 1979, 17, 1103.
- Chu, D. Y.; Thomas, J. K. Macromolecules 1984, 17, 2142. (12) Thomas, J. K. The Chemistry of Excitation of Interfaces; American Chemical Society: Washington, DC, 1984; p 283.
- (13) Pyrene was dispersed in 1 mg/mL PEAA solutions in the required phosphate buffers (note 4) with prolonged stirring at room temperature. Emission spectra were recorded at room temperature on a Perkin-Elmer MPF-66 spectrometer.
- (14) Kalvanasundaram, K.: Thomas, J. K. J. Am. Chem. Soc. 1977, 99, 2039.
- (15) The copolymer was prepared by bulk radical copolymerization of 2-ethylacrylic acid and 1-pyreneacrylic acid, initiated by azobis(isobutyronitrile) at 60 °C. The copolymer was purified by three dissolution-precipitation cycles, with N.N-dimethylformamide as solvent and ethyl acetate as nonsolvent. The copolymer contained 1.5×10^{-3} mol % 1-pyreneacrylic acid, as determined from the ultraviolet absorption spectrum. The inherent viscosity of the copolymer (0.2% in DMF, 35 °C) was $0.63 \, dL/g$.
- Mixing of the copolymer and DPPC was accomplished in the following manner: 0.9 mL of a 1 mg/mL copolymer solution in the required phosphate buffer (note 4) was mixed at room temperature with 0.1 mL of a 10 mg/mL suspension of DPPC in the same buffer. The mixture was then heated to 50 °C for 10 min with repeated vortex agitation.
- (17) Tsuchida, E.; Abe, K. Adv. Polym. Sci. 1982, 45, 1.
- (18) Borden, K. A.; Tan, J. S.; Tirrell, D. A., unpublished results. (19) Liquori, A. M.; Barone, G.; Crescenzi, V.; Quadrifoglio, F.; Vitagliano, V. J. Macromol. Chem. 1966, 1, 291.
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Identification of a Curie Transition in Vinylidene Fluoride/Tetrafluoroethylene Random Copolymers by Spectroscopic Methods

While copolymers of vinylidene fluoride (VF2) and trifluoroethylene (TrFE) have been studied extensively, 1-6 only recently⁷⁻⁹ has a Curie transition been observed in random copolymers of VF₂ and tetrafluoroethylene (TFE). Initial studies on a commercially available 81/19 (mol %) copolymer had mixed results^{7,8,10-12} in detecting an observable Curie transition due to its proximity to the crystalline melting point. However, a systematic study^{13,14} of VF₂/TFE copolymers over a wide compositional range by X-ray and thermal analysis unequivocally demonstrated the existence of a Curie transition.

Since the random addition of TFE comonomer to a VF₂ backbone does not introduce a new chemical species (as in the case of copolymerization with TrFE) but merely increases the head-to-head defect content, evidence of a Curie transition in VF₂/TFE copolymers further reinforces the notion that poly(vinylidene fluoride) (PVF₂) must also undergo a Curie transition.7 The fact that it may coincide with the crystalline melting point remains a plausible argument for the lack of any experimental verification of this Curie point in PVF₂. In the commercially available (Kynar 7201) VF₂/TFE (81/19) copolymer, a similar observation can be made, as shown in Figure 1. On the first

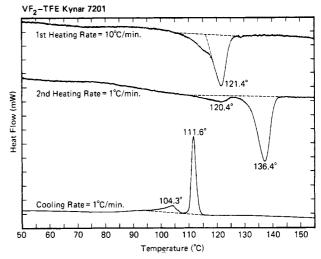


Figure 1. DSC trace of VF₂/TFE copolymer with a 81/19 (mol %) ratio. Top trace, first heating; lower trace, cooling at 1 °C/min; middle trace, sample cooled from the melt at 0.2 °C/min and then heated.

heating the DSC trace shows an asymmetric melting point curve indicating the presence of a premelting transition at 116 °C. When the sample is cooled (bottom trace), it becomes quite evident that in addition to the crystallization exotherm, there is a second, weaker feature, which might be associated with a paraelectric to ferroelectric phase transition.

As earlier work⁶ indicated, Raman spectroscopy is a nondestructive technique for probing local changes in conformational order which accompany a Curie transition. In fact, bands in the 800-850-cm⁻¹ region have been used⁶ to investigate the loss of the planar PVF₂ structure in VF₂/TrFE copolymers as the temperature was raised to the Curie point. The intense 841-cm⁻¹ asymmetric CF₂ stretching band which is characteristic of the planar form was observed to lose intensity as the 805-cm⁻¹ band characteristic of TG and/or TG' conformations became dominant at the Curie point. It was thus anticipated that similar high-temperature studies could be carried out on the VF₂/TFE copolymers and could be used to establish the absence or existence of a Curie transition.

In order to investigate the transition shown in Figure 1 by Raman spectroscopy, a sample of the Kynar 7201 copolymer was slow cooled from the melt at 0.2 °C/min. This allowed more perfect crystals to form, thereby raising the melting point by 15 °C as shown in the second heating trace of Figure 1. Moreover, of equal importance was the appearance of a well-defined transition at 120.4 °C, easily delineated from the melting point at 136.4 °C. A second portion of this same sample was used for the Raman studies shown in Figure 2. At 83 °C the spectrum is very similar to that observed⁶ for the VF₂/TrFE copolymers below the Curie point. As the temperature approaches that of the observed DSC transition (120.4 °C) a dramatic change in the intensity of the bands at 805 and 841 cm⁻¹ is observed, with the 805-cm⁻¹ band increasing in intensity at the expense of the 841-cm⁻¹ band. Lowering the temperature to 108 °C returns the relative intensity of the bands to their original value. The cyclic reversibility and the band intensity variations with temperature suggest that the 120.4 °C transition involves a reversible change in conformation from a planar structure to one of lesser order which includes short TGTG' sequences. This order-disorder transition is thus identical with that observed⁶ in VF₂/TrFE copolymers and can be characterized as a Curie transition.

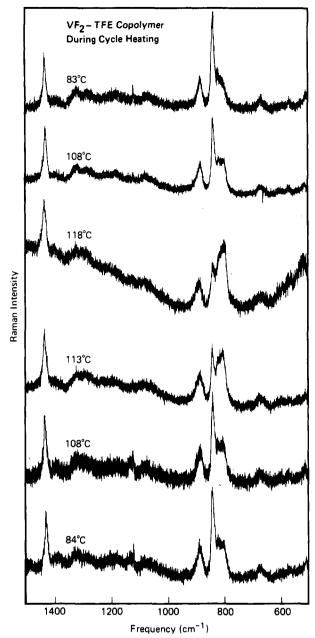


Figure 2. Raman spectra of VF₂/TFE copolymer as a function of temperature. (excitation wavelength = 488.0 nm, resolution $= 4 \text{ cm}^{-1}$).

The feasibility of using Raman measurements to determine Curie behavior in VF2 copolymers has been demonstrated. The characterization of these order-disorder transitions by nondestructive spectroscopic methods will be of extreme importance in studying VF2/TFE copolymers with VF₂ compositions¹⁵ less than 50%. In these cases the Curie point will occur below room temperature and will not be easily accessible by conventional X-ray diffraction methods.

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Registry No. Kynar 7201, 25684-76-8.

References and Notes

- Yagi, T.; Tatemoto, M.; Sako, J. Polym. J. (Tokyo) 1980, 12, (1) 209.
- Furukawa, T.; Johnson, G. E.; Bair, H. E.; Tajitsu, Y.; Chiba, A.; Fukada, E. Ferroelectrics 1981, 32, 61.

- (3) Yamada, T.; Ueda, T.; Kitayama, T. J. Appl. Phys. 1981, 52, 948.
- Tashiro, K.; Takano, K.; Kobayashi, M.; Chatani, Y.; Tado-
- koro, H. Polymer 1981, 22, 1312. Lovinger, A. J.; Davis, G. T.; Furukawa, T.; Broadhurst, M. G. Macromolecules 1982, 15, 323.
- Green, J. S.; Rabe, J. P.; Rabolt, J. F. Macromolecules 1986,
- Lovinger, A. J. Macromolecules 1983, 16, 1529.
- Lovinger, A. J.; Johnson, G. E.; Bair, H. E.; Anderson, E. W. J. Appl. Phys. 1984, 56, 2412.
- Leonard, C.; Halary, J. L.; Monnerie, L.; Micheron, F. Polym. Bull. (Berlin) 1984, 11, 195.
- (10) Hicks, J. C.; Jones, T. E.; Logan, J. C. J. Appl. Phys. 1978, 49, 6092.
- (11) Koizumi, N.; Hagino, J.; Murata, Y. Ferroelectrics 1981, 32, 141.
- Tasaka, S.; Miyata, S. J. Appl. Phys. 1985, 57, 906.
- Lovinger, A. J.; Davis, D. D.; Cais, R. E.; Kometani, J. M. Macromolecules 1986, 19, 1491.
- Murata, Y.; Koizumi, N. Polym. J. (Tokyo) 1985, 17, 1071. Lovinger, A. J.; Cais, R. E.; Bair, H. E.; Johnson, G. E.; Davis, D. D.; Kometani, J. M.; Anderson, E. W. Bull. Am. Phys. Soc. 1986, 31, 611.

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Direct Measurement of Propagating Radical Concentration in a Semicontinuous Emulsion Polymerization

Achievement of a better fundamental understanding of free radical emulsion polymerization requires knowledge of the nature and concentration of the propagating free radical species. Direct observation of propagating free radicals in an emulsion polymerization had not been reported until the recent work of Ballard et al., 1,2 who applied modern ESR techniques to observe propagating free radicals in a batch emulsion polymerization of methyl methacrylate. We have extended ESR analysis to semicontinuous acrylic emulsion polymerization and have studied the propagating free radical species as a function of polymerization temperature and particle size.

Semicontinuous emulsion polymerization, in which monomer is continuously added over several hours, is commonly used for the commercial preparation of latex polymers. This method is preferred primarily because it allows easy control of the polymerization temperature and the preparation of uniform copolymer compositions. Under suitable conditions the rate of polymerization is governed by the monomer feed rate. If it is assumed that monomer diffusion is not a rate-limiting process and that the bulk of the polymerization occurs isotropically within the latex particles, then in a semicontinuous system that has reached steady state the polymerization kinetics can be represented by eq 1, where [M] and [R] represent the

rate of monomer feed = rate of polymerization =

 $k_{\rm p}[{\rm M}][{\rm R}^{\bullet}]$ (1)

steady-state monomer and radical concentrations in the latex particles and k_p is the propagation rate constant. The validity of this equation has not been previously tested by independently measuring each term. The rate of monomer feed and the steady-state monomer concentration can be measured experimentally, and k_p can be estimated from analogous homogeneous polymerizations. The ESR measurements reported here now give us the steady-state radical concentration data which show that eq 1 is valid